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## Section 5. Thermophysical properties and the thermo-electric effect

# Thermal properties of Mo<sub>3</sub>Te<sub>4</sub>

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#### Abstract

The thermal properties of  $Mo_3Te_4$  were investigated in the temperature range from 300 to 1000 K. The linear thermal expansion coefficient of  $Mo_3Te_4$  was evaluated from high-temperature X-ray diffraction. The thermal conductivity was calculated from the heat capacity, the pycnometric density, and the thermal diffusivity measured by the laser flash method. The thermal conductivity of  $Mo_3Te_4$  is relatively low and is increased from 2.2 W m<sup>-1</sup> K<sup>-1</sup> at 300 K to 3.1 W m<sup>-1</sup> K<sup>-1</sup> at 1000 K. The electrical conductivity was measured by the standard four-probe dc method. The electronic contribution to the thermal conductivity was calculated using the Wiedemann–Franz law, and the lattice thermal conductivity was estimated. The mechanism of the thermal conduction of  $Mo_3Te_4$  was discussed. © 2001 Elsevier Science B.V. All rights reserved.

#### 1. Introduction

Molybdenum and tellurium have comparatively high fission yields, and these elements are the predominant fission products in several types of metallic precipitates of spent fast reactor fuel [1,2]. In order to evaluate the safety of the fuel, it is important to understand the effect of the fission products on thermal and mechanical properties of the fuel under irradiation. However, the thermal properties on the molybdenum telluride intermetallic compounds have scarcely reported. In the present study, several thermal properties such as the linear thermal expansion coefficient and the thermal conductivity of  $Mo_3Te_4$  were investigated in the temperature range from 300 to 1000 K.

### 2. Experimental

Mo<sub>3</sub>Te<sub>4</sub> was prepared by mixing stoichiometric amounts of the elements followed by heating in sealed quartz ampoules for 2 days at 1023 K. The powder was then hot-pressed in graphite dies into dense samples. The

hot-pressing was conducted at a pressure of about 8 MPa and at the temperature 1273 K for 2 h under argon atmosphere. The crystal structure of the sample was analyzed by the powder X-ray diffraction method at room temperature using Cu– $K_{\alpha}$  radiation. For thermal and electrical properties measurements, appropriate shapes of the samples were cut from the pellet. The density of the samples was calculated from the measured weight and dimensions.

The linear thermal expansion coefficient of Mo<sub>3</sub>Te<sub>4</sub> was evaluated by the high temperature X-ray diffraction method in the temperature range from room temperature to about 700 K under argon atmosphere. Between room temperature and about 1200 K, the thermal conductivity of Mo<sub>3</sub>Te<sub>4</sub> was calculated from the heat capacity [3], the pycnometric density, and the thermal diffusivity measured by the laser flash method using ULVAC TC-7000 in vacuum. The electrical conductivity was measured by using the standard four-probe dc method under helium atmosphere between room temperature and about 923 K.

### 3. Results and discussion

The powder X-ray diffraction pattern of the sample is shown in Fig. 1 together with the literature data [4]. This figure shows that single-phase Mo<sub>3</sub>Te<sub>4</sub> with a hexagonal structure (Chevrel phase) is obtained in the present

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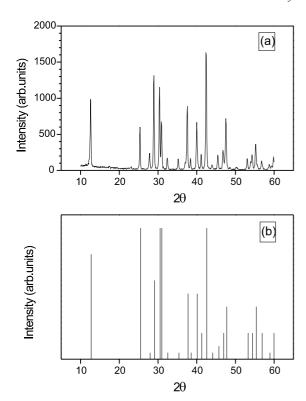


Fig. 1. (a) X-ray powder diffraction pattern for Mo<sub>3</sub>Te<sub>4</sub>; (b) simulated X-ray powder diffraction pattern for Mo<sub>3</sub>Te<sub>4</sub> [4].

study. The lattice parameters and X-ray density of Mo<sub>3</sub>Te<sub>4</sub> were obtained from the X-ray diffraction analysis. The bulk densities of the samples are about 80% of the theoretical density. Although the Mo/Te ratio was not measured in the present study, the deviation from the stoichiometry is small. This is because the lattice parameters of Mo<sub>3</sub>Te<sub>4</sub> prepared in the present study are close to the previously reported literature data [5–7]. The sample characterizations of Mo<sub>3</sub>Te<sub>4</sub> are shown in Table 1.

Fig. 2 shows the temperature dependence of the hexagonal unit cell parameters a, c and V of Mo<sub>3</sub>Te<sub>4</sub>. All parameters increase smoothly with increasing temperature. These temperature dependences can be expressed as a linear function of temperature. Between room temperature and about 623 K the linear and volumetric thermal expansion coefficients  $(\alpha_a, \alpha_c, \alpha_V)$  were calculated from change in the lattice parameters and lattice volume with temperature by using the following equation:

$$\alpha_a = \frac{1}{a_0} \left( \frac{\partial a(T)}{\partial T} \right)_P,\tag{1}$$

$$\alpha_c = \frac{1}{c_0} \left( \frac{\partial c(T)}{\partial T} \right)_P, \tag{2}$$

Table 1 Lattice parameters at room temperature, pycnometric and X-ray densities, and linear and volumetric thermal expansion coefficient between room temperature and 623 K for Mo<sub>3</sub>Te<sub>4</sub>

Lattice parameters (at	
room temperature) (nm)	
a	1.0179
c	1.1725
X-ray density (Mg/m <sup>3</sup> )	7.6
Bulk density	
$(Mg/m^3)$	6.2
(%T.D.)	82
Linear thermal expansion	
coefficient $(K^{-1})$	
$\alpha_a$	$1.58 \times 10^{-5}$
$\alpha_c$	$8.25 \times 10^{-6}$
Volumetric thermal	
expansion coefficient (K <sup>-1</sup> )	
$lpha_V$	$4.04 \times 10^{-5}$

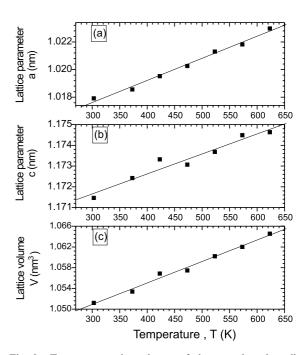


Fig. 2. Temperature dependence of hexagonal unit cell parameters a, c and lattice volume V for  $Mo_3Te_4$ .

$$\alpha_V = \frac{1}{V_0} \left( \frac{\partial V(T)}{\partial T} \right)_P,\tag{3}$$

where  $a_0, c_0$  and  $V_0$  are the lattice parameters at room temperature. These results are shown in Table 1. It is found that the linear thermal expansion coefficient in a direction  $\alpha_a$  is larger than that in c direction  $\alpha_c$ .

It is empirically found that the product of  $\alpha_L$  and  $T_{\rm m}$  is a constant for many substances, where  $\alpha_L$  equals one-third of  $\alpha_V$  and  $T_{\rm m}$  is the melting temperature in K [8,9].

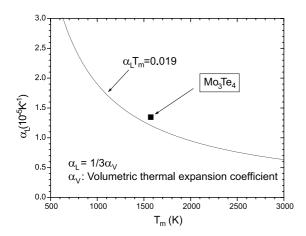


Fig. 3. Relationship between  $\alpha_L$  and  $T_m$  for Mo<sub>3</sub>Te<sub>4</sub>.

For pure metals the following relationship between  $\alpha_L$  and  $T_m$  was reported [8,9]:

$$\alpha_L T_{\rm m} = 0.019. \tag{4}$$

The variation of  $\alpha_L$  with  $T_{\rm m}$  [10,11] are shown in Fig. 3. It is found from this figure that the relationship between  $\alpha_L$  and  $T_{\rm m}$  of Mo<sub>3</sub>Te<sub>4</sub> has the same tendency as for pure metals.

The thermal conductivity  $\lambda$  was calculated from the measured thermal diffusivity a, specific heat capacity  $c_p$ , and density d using the following relationship:

$$\lambda = ac_{p}d. \tag{5}$$

The temperature dependence of the thermal conductivity of Mo<sub>3</sub>Te<sub>4</sub> is shown in Fig. 4. These experimental data were corrected to 100% T.D. by using Schulz's equation [12]. At around room temperature the thermal conductivity of Mo<sub>3</sub>Te<sub>4</sub> is extremely lower than that of UO<sub>2</sub>

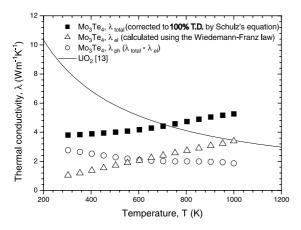


Fig. 4. Temperature dependence of thermal conductivity for Mo<sub>3</sub>Te<sub>4</sub>.

[13], while it is increased gradually with increasing temperature, and becomes higher than that of  $UO_2$ . This temperature dependence shows metallic behavior.

It is well known that the total thermal conductivity of the material  $\lambda_{total}$  of solids can be written as

$$\lambda_{\text{total}} = \lambda_{\text{el}} + \lambda_{\text{ph}},$$
 (6)

where  $\lambda_{el}$  is the electronic contribution and  $\lambda_{ph}$  is the lattice contribution. In order to separate  $\lambda_{ph}$  from  $\lambda_{el}$ , the electronic contribution to the thermal conductivity was calculated using the Wiedemann–Franz law,

$$\lambda_{\rm el} = L\sigma T,\tag{7}$$

where L is the Lorentz number,  $\sigma$  is the electrical conductivity, and T is the absolute temperature. In the present study, the electrical conductivity of Mo<sub>3</sub>Te<sub>4</sub> was measured by using the standard four-probe dc method in the temperature range from room temperature to about 1000 K, and this result is shown in Fig. 5. The electrical conductivity of Mo<sub>3</sub>Te<sub>4</sub> are almost independent of the temperature, and the values are about  $1.4 \times 10^5 \ \Omega^{-1} \ m^{-1}$ . The lattice contribution was obtained by subtracting  $\lambda_{el}$  from  $\lambda_{total}$ . Variations of  $\lambda_{ph}$ ,  $\lambda_{el}$ , and  $\lambda_{total}$  with temperature are shown in Fig. 4. The temperature dependence of  $\lambda_{ph}$  follows a 1/T law, which indicates that the  $\lambda_{ph}$  is composed of phonon contribution. In the low temperature range, the lattice contribution is more predominant than the electronic contribution, while in the high temperature range the electronic contribution increases rapidly and it becomes comparable to the lattice contribution. From the present study, it is found that the characteristics of the thermal properties such as the thermal expansion, melting point, and thermal conductivity of Mo<sub>3</sub>Te<sub>4</sub> show metallic behavior.

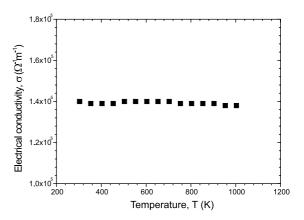


Fig. 5. Temperature dependence of electrical conductivity for  $Mo_3Te_4$ .

#### 4. Conclusion

The thermal properties of Mo<sub>3</sub>Te<sub>4</sub> were investigated in the temperature range from 300 to 1000 K. The hexagonal unit cell parameters increased smoothly with increasing temperature. The relationship between the thermal expansion coefficient and the melting temperature of Mo<sub>3</sub>Te<sub>4</sub> had the same tendency as for pure metals. The thermal conductivity of Mo<sub>3</sub>Te<sub>4</sub> increased with increasing temperature, indicating metallic behavior. The electrical conductivity of Mo<sub>3</sub>Te<sub>4</sub> was measured by using the standard four-probe dc method, and the electronic contribution to the thermal conductivity was evaluated by using the Wiedemann-Franz law. In the low temperature range, the lattice contribution to the thermal conductivity was more predominant than the electronic contribution, while in the high temperature range the electronic contribution became comparable to the lattice contribution.

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